กระแสวิทยาของกราฟืนสังเคราะห์ที่แขวนลอยในพอลิฟินิลไกลซิคิลอีเทอร์โคฟอล์มาลดีไฮค์

นาย พลวัตน์ เจริญธรขจรชัย

วิทยานิพนธ์นี้เป็นส่วนหนึ่งของการศึกษาตามหลักสูตรปริญญาวิศวกรรมศาสตรมหาบัณฑิต สาขาวิชาวิศวกรรมเคมี ภาควิชาวิศวกรรมเคมี คณะวิศวกรรมศาสตร์ จุฬาลงกรณ์มหาวิทยาลัย ปีการศึกษา 2554 ลิขสิทธิ์ของจุฬาลงกรณ์มหาวิทยาลัย

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RHEOLOGY OF SYNTHESIZED GRAPHENE SUSPENDED IN POLY[(PHENYL GLYCIDYL ETHER)-CO-FORMALDEHYDE]

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A Thesis Submitted in Partial Fulfillment of the Requirements for the Degree of Master of Engineering Program in Chemical Engineering Department of Chemical Engineering Faculty of Engineering Chulalongkorn University Academic Year 2011 Copyright of Chulalongkorn University

Thesis Title	RHEOLOGY OF SYNTHESIZED GRAPHENE SUSPENDED IN
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กราฟีนถูกสังเคราะห์จากกราไฟต์ที่มีความบริสทธิ์สูงผ่านสองกระบวนการ คือ กระบวนการสังเคราะห์กราฟีนออกไซด์ด้วยวิธีการของฮัมเมอร์ที่ปรับปรุงแล้วและวิธีการกำจัดหมู่ ้ออกซิเจนโดยใช้ไฮคราซีน โดยกราฟีนออกไซด์จะถูกปรับก่าความเป็นกรค-ค่างของสารแขวนลอย ในน้ำให้เหมาะสมก่อน เนื่องจากกราฟีนออกไซด์ยังแขวนลอยในน้ำที่สภาวะความเป็นกลางได้ต่ำ จากการวิจัยพบว่าที่ค่าความเป็นด่างเท่ากับ12 เป็นสภาวะที่เหมาะสมที่สุด จากนั้นจึงทำการกำจัด หมู่ออกซิเจนออกโดยใช้ไฮคราซีน โดยเสถียรภาพของคอถลอยค์จะถูกวิเคราะห์ด้วยเกรื่อง UV-VIS spectroscopy และวัดปริมาณประจุบนผิวของอนุภาคที่เกิดจากการปรับค่าความเป็นกรด-ด่าง ้ด้วยวิธี zeta-potential อนุภาคที่ได้ในแต่ละขั้นตอนจะถูกนำมาวิเคราะห์สมบัติทางกายภาพโดยใช้ Raman spectroscopy, SEM, TEM และ AFM ในขณะที่การวิเคราะห์หมู่ฟังก์ชั่นและสมบัติทาง ้ความร้อนของอนุภาคถูกวิเคราะห์ด้วยเครื่อง FTIR และ TG/DTA ตามลำดับ จากนั้นนำกราฟืนที่ ผสมลงในอีพอกซี, พอลิฟีนิลไกลซิคิลอีเทอร์ โคฟอร์มาลดีไฮค์ โคยใช้cone and plate rheometer ที่ 1-3 % โดยน้ำหนักมาวิเคราะห์สมบัติการ ใหลที่สภาวะ steady state และที่ 5-9 % โดยน้ำหนักของ กราฟืนวิเคราะห์สมบัติการใหลจาก oscillatory measurement โดยพบว่าสมบัติการใหลสามารถ อธิบายได้ด้วยโมเคลของ Casson และสมบัติความแข็งและไหลได้สามารถอธิบายได้จากวิธีของการ เปลี่ยนแปลงความเครียดและการเปลี่ยนแปลงความถึ่

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POLLAWAT JAROENTHONKAJONCHAI: RHEOLOGY OF SYNTHESIZED GRAPHENE SUSPENDED IN POLY[(PHENYL GLYCIDYL ETHER)-CO-FORMALDEHYDE]. ADVISOR: ASST.PROF.ANONGNAT SOMWANGTHANAROJ,Ph.D., 68 pp.

Graphene was synthesized through two-step method. In the first step, graphene oxide (GO) was synthesized from purified graphite via modified Hummers method. Because graphene oxide poorly dispersed in aqueous solution at pH 7, the surface treatment by adjusting pH of solution for repulsive charge and homogeneous colloid solution before reducing GO with hydrazine was studied in this research. The colloidal stability of graphene oxide in aqueous solution was measured by UV-VIS spectroscopy in term of spectrum absorption and the zeta- potential technique was used for detecting surface change of particle. Then, the chemical reduction of graphene oxide to graphene sheet was done by using hydrazine hydrate. The physical properties of particle were characterized by Raman spectroscopy, SEM, TEM and AFM technique. The functional group and thermal stability of particle were characterized via FTIR, TG/DTA respectively. Graphene sheet was then mixed with epoxy, poly[(phenyl glycidyl ether)-co-formaldehyde] at 1-3 wt% loading for steady state measurement and 5-9 wt% loading for oscillatory measurement. The rheological properties of epoxy-graphene suspension were studied by cone and plate fixture. The rheological flow model was fitted in Casson model. The solid and liquid like behavior of suspension was detected from strain sweep and frequency sweep technique.

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V

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CHAPTER I

INTRODUCTION

1.1 General Introduction

In recent y ears, gr aphene was d iscovered and the r esearchers found that graphene e xhibits high electrical co nductivity and thermal s tability t hus it is interesting m aterial for using i n s everal applications such as sensor and electronic devices. The structure of gr aphene is sp² carbon i n hexagonal ho neycomb s tructure bonding interlayer with V an der W aals force. Although there are several methods to synthesize graphene, t he synthesized products have many s tructures a nd various numbers of layers depending on method and preparation process. Graphene oxide is the plate-like colloid when suspended in water which is less soluble at pH 7 and the stability in water is one thing researchers would like to improve. In general way for improving dispersion and stability of graphene in water is to treat charge repulsion on surface, using steric hindrance behavior from adding either polymer or surfactant and using both o ft heir methods by polyelectrolyte. T he most suitable a nd easy preparation method is used in this research.

Because of t heir e xcellent pr operties, gr aphene is t he interesting filler in polymer. The rheological properties are the important information to study the flow properties of gr aphene d ispersion w hen app lied s hear force w hich is useful f or polymer composite pr ocessing. The rheology is a us eful method for s tudying of particle interaction a nd structures i n suspension o f co lloidal particle. G raphene suspension is t he c olloidal system in which information in r heology is still limited thus we would like to explore it.

In this research, graphene o xide was prepared by chemical method because this method can produce high yield of products from synthesis and easy preparation. The p H of solution was then a djusted to obtain repulsive surface charge. Graphene oxide (GO) was reduced with hydrazine. After that, graphene product was mixed with epoxy, po ly [(phenyl g lycidyl e ther)-co-formaldehyde] by s onication. Rheological properties of graphene suspended in epoxy via cone-and-plate rheometer were studied by varying concentration of graphene.

1.2 Objectives

- 1. To synthesize graphene from graphite via chemical method
- 2. To study stability of graphene oxide in aqueous suspension when varying pH
- 3. To ob tain rheological pr operties o f gr aphene suspended in po ly [(phenyl glycidyl ether)-co-formaldehyde]

1.3 Scopes of the research

- 1. Graphene oxide is prepared from graphite by modified Hummers method.
- Graphene is prepared from graphene oxide via chemical reduction technique with hydrazine hydrate.
- 3. The surface charge of gr aphene o xide in aqueous suspension whose z etapotential is measured is varied by HCl and NaOH at pH 3-14.

- 4. Morphology, t hermal s tability, an d chemical structure of gr aphene ar e characterized.
- 5. The flow behavior of 0-9 wt% graphene suspended in poly [(phenyl glycidyl ether)-co-formaldehyde] are studied by m eans o fr heological t echnique (0-3 wt% for steady state and 5-9 wt% for oscillation measurement).

CHAPTER II

LITERATURE REVIEWS

2.1 Graphene

Graphene is an allotrope of carbon atom packing in a form of two dimensional hexagonal h oneycomb lattices. It can be produced from hy drocarbon materials, for example, graphite, carbon nanotube and hydrocarbon gases such as methane, ethylene and benzene. Since graphene was discovered by Geim and co-workers in 2004 us ing scotch tape method, many investigations have been reported its excellent properties for utilizing in versatile applications. Many researchers found that graphene showed excellent electrical, thermal and mechanical properties. [1-5] Graphene is classified as a s emiconductor w hich shows hi gh e lectrical mobility (200,000 c m²V⁻¹s⁻¹) a nd electrical co nductivity is slightly higher t han t hat of c arbon nanotube. M oreover, graphene po ssesses superior thermal conductivity a bout 4,000-5,000 W m⁻¹K⁻¹ at r oom temperature. In consideration of mechanical property, tensile strength of graphene is about 130 GPa, which is about 100 times greater than nanosteel and Young's modulus is about 1,100 GPa.

The t echniques t o s ynthesize gr aphene have been de veloped rapidly. T he important factors for s election t he process for graphene s ynthesis are high quality

(such as number of graphene layer), high quantity and easy procedure. The summary of methods, pr ocedure of gr aphene synthesis a nd number of graphene layer in products a re de picted in T able 2.1 [1, 2, 6-10]. The number of graphene layers is difficult t o ch aracterize because gr aphene layer is very t hin. It m ust b e measured using more t han one method for c omparison. In general, the t hickness of gr aphene layer ca n be measured by microscopy t echniques s uch a st ransmission e lectron microscopy (TEM) and atomic force microscopy (AFM) and it can also be identified by Raman spectroscopy.

Micromechanical cleavage or pe el o ff method [1] produces s ingle-layer graphene but i t i s h ard t o s cale-up b ecause o f v ery low pr oductivity. Although chemical vapor de position method (CVD) [6] generates s ingle-layer gr aphene, removal o f gr aphene from metal s ubstrate i s difficult a nd high t emperature i s required. Thermal desorption of silicon or epitaxial growth method also gives singlelayer gr aphene but expensive substrate such as 6 H-SiC and 4H-SiC i n high thermal process is r equired. D espite de veloping s ubstrate w ith nickel t hin film, t he temperature in process is still higher than 700 °C. Monolayer of graphene is produced from electrochemical method [5] using graphite rod as the electrode but this method is hard to scale-up and operate in industrial scale since skillful expert is required for this technique.

Several r esearch gr oups continually de veloped the t echnique t o s ynthesize few-layer (2-10 layers) gr aphene. S ynthesis from nanodiamond by heating graphite furnace is not appropriate because it requires very high temperature (2,200°C) which is difficult for operation and scaling-up. Arc-discharge evaporation using graphite rod and electrode i s qu ite complicated and special expert is needed for o peration. I n comparison to above-mentioned synthesis methods, chemical method is chosen in this project because of low temperature process, simple for preparing equipment and easy to scale-up to industrial process. Chemical method can be divided into two main parts: conversion of graphite powder to graphene oxide and reduction of graphene oxide to graphene.

Two we ll-known c hemical methods for c onversion of graphite po wder t o graphene oxide (GO), in which strong oxidant is used for reaction, are Staudenmaier method and H ummers method. I n S taudenmaier method, co ncentrated nitric a nd sulfuric acid is reacted with potassium chlorate to produce chlorine dioxide gas which is the toxic gas and can explode if concentration of chlorine dioxide gas is too high. Thus, Hummers method is c hosen for t his w ork. S imilar t o Staudenmaier method, concentrated sulfuric acid is us ed but o xidizing a gent is c hanged from po tassium chlorate t o potassium p ermanganate w hich is much safer. [2] Graphene o xide s heet from t his pr ocess is e xfoliated from gr aphite a nd separated to thin layers. After oxidation, gr aphene o xide sheet c onsists o f e poxide and hydroxyl gr oup on b asal plane, as well as carbonyl and carboxyl group at the edge of sheet as shown in Figure 2.1.

 Table 2.1 Methods of graphene synthesis

Methods	Ref	No of layers	Procedure	
Micromechanical	[1]	1	Peel off graphene from highly ordered graphite by using scotch tape or adhesive tab	
cleavage			and transfer to Si substrate	
Thermal			Use 6H-SiC to insert in electron bombardment under vacuum at 100 °C, then heat up	
desorption or	[1, 6]	1	to 1250-1450 °C and cool down on Ni substrate or using Ni coat on substrate and heat	
epitaxial growth			up	
Chemical vapor	[1 11]	1	Flow hydrocarbon gas that dilute with Ar under H_2 through substrate which is coated	
deposition (CVD)	[1, 11]		by metal film at 800-1000 °C	
Electrochemical	[6]	1	Using purified graphite rod as electrode and apply static potential in ionic liquid-	
modification	[3]		water mixture	
Arc-discharge	[1]	2-3	Using purified graphite rod as anode and another graphite rod as cathode and apply	
evaporation			current under H_2 and He in water chamber	
Chemical			Mixed graphite powder with concentrated acid and oxidizing agent in ice bath and	
Reduction with	[1, 7, 8]	2-6	then heat up to 35 $^{\circ}$ C in 2 hours, add DI water and H ₂ O ₂ until color change to yellow	
reducing agent			brown, wash until neutral and reduced with hydrazine in reflux	
Thermal	[2]	2-7	The process to synthesize GO is the same as chemical reduction but using GO putting	
exfoliation of GO			in long quartz tube and insert to furnace for reducing oxygen functional group	
Conversion of	[1, 9]	4-8	Insert nanodiamond powder in graphite furnace under He or vacuum at 1650-2200 °C	
nanodiamond	L / J			



Figure 2.1 The illustration of structure of graphene oxide. [2]

The t hickness of e xfoliated gr aphene o xide s heet i n a queous s uspension w as measured by at omic force s pectroscopy. The result w as 1 nm thickness indicating that GO was slightly thicker than 0.34 nm thickness of pristine graphene sheet which might be due t o the e ffect of c ovalent bond and oxygen functional group on ba sal p lane of graphene surface [8].

However, graphene o xide can be dispersed in water or organic solvent at only small concentration and it tends to agglomerate at high concentration. Thus, it restricts the app lication o f gr aphene o xide t hat r equires high c oncentration. G raphene o xide suspension is the colloidal system; therefore, the stability o f gr aphene o xide s hould be improved to produce well dispersion as depicted below.

2.2 Stability of graphene oxide and graphene

Graphene o xide and graphene a re in the colloidal system when suspended in water and organic solvent. Although they can be dispersed in water and organic solvent better than graphite, only a small amount of them is dispersed. From this reason, many researchers have tried to find method for stabilization. In general, stability is improved by using e lectrostatic r epulsive or s urface c harge method, s teric hindrance from covalent linkage, combination between surface charge and steric hindrance such as polyelectrolyte and physical adsorption which are shown in Figure 2.2. [12]



Figure 2.2 The stabilization methods of graphene in aqueous suspension a) the introduction of negative charges, b) the formation of covalent linkages, c) π - π interaction of (poly)electrolytes, and d) hydrophobic interaction of non-(poly)electrolytes. [12]

Modification of functional group by bonding covalent linkage between graphene sheet and lyophilic or conducting polymer via polymerization have been studied but this method reduced electrical conductivity of graphene sheet because of a very low electrical conductivity of lyophilic and conducting polymer. [13-15] The electrostatic repulsive of graphene and graphene oxide in aqueous suspension at various pH is the low-cost method without af fecting e lectrical co nductivity because it do es n ot b ond with polymer or surfactant. The electrostatic stability of graphene sheet depends on pH value, electrolyte concentration and concentration of p article d ispersion. The graphene d ispersion c an be stable if the electrostatic repulsive force is greater than van der W aals force between particles.

The electrostatic stability of this system is measured by zeta-potential technique. This technique measures surface charge of particle. If there are many of surface charges of particle, the zeta-potential is far away from zero. The point of zero surface charge of particle is ca lled isoelectric po int where t he particle is ea sy t o aggl omerate. The isoelectric point is the specific property of particle, for example, titania has isoelectric point around pH 5.8. The zeta-potential is the representative of surface charge potential which is measured between s tern l ayer and diffuse l ayer. The surface c harge v alue reduces e xponentially until zero from particle to surface. This d istance layer is ca lled electrical do uble layer or D ebye length. [16] Figure 2.3 shows the stern layer, d iffuse layer and surface charge character of particle. If the zeta-potential is greater than + 30 or less than -30 mV, it s hows the stable colloidal system. [17] From this basic colloidal system, pr eparation of gr aphene a nd gr aphene o xide in a queous suspension us ing

electrostatic s tabilization at v arious pH values is t he interesting technique. For electrostatic s tabilization of particle in suspension, t he colloidal system is stable if t he electrostatic repulsive charge is greater than van der Waals force. Although this technique was st udied by using a mmonia [17] and po tassium hydroxide [18], there ar e s ome different conditions during synthesis. Hence, the surface charge potential in our case may be different from their researches.



Figure 2.3 Behavior of surface charge particle via zeta-potential technique [16]

Even though oxygen functional group (carboxylic, hydroxyl, and epoxide) that occurred from conversion of graphite powder to graphene oxide process can form more stable colloidal suspension in water than graphene sheet, this oxygen functional group must be removed because it reduces electrical conductivity of pristine graphene sheet.

2.3 Chemical reduction of graphene oxide

There are two processes for removing oxygen on graphene oxide sheet. The first one is thermal process that is called thermal exfoliation of graphene oxide. This method begins with putting graphene oxide powder into a long quartz tube and inserting it in tube furnace which is heated up to 1,050°C. The thermal process breaks the functional group of oxygen on graphene surface [1, 2]. This method must use the specifically designed equipment and high t emperature. A nother method is c hemical r eduction by u sing reducing agent. The commonly used reducing agent is hydrazine. Although hydrazine is a toxic substance and can be exploded if the minimum concentration reaches, it is still the best reducing agent for chemical reduction of graphene oxide to graphene at this time in comparison to other reducing agents such as ethylene glycol which is difficult to handle. [1, 8]

There are three factors that affect the process for reducing graphene via chemical reduction with hydrazine that are temperature, ratio of hydrazine to graphene oxide and time of reaction. They a ffect how much oxygen can be removed from graphene sheet. Graphene sheet ha s low o xygen o n s urface if high t emperature, hi gh hydrazine concentration and long reaction time are used. However, the reaction temperature is the crucial factor that must be chosen carefully and it should not be too high to a void the explosion because hydrazine has flash point around 96°C. Furthermore, the reaction must be o perated i n c losed system u nder r eflux co ndition t o c ondense t he ev aporated hydrazine. Li and coworker [17] studied the influence of ratio of hydrazine to graphene oxide on electrical conductivity and they found that 7: 10 hydrazine to graphene oxide

ratio gave the electrical conductivity about 7,000 S.m^{-1} and it do es not change when the amount o f hy drazine further increases as c onfirmed by e lemental a nalysis as demonstrated in Table 2.2.

	C	N	Н	0			Conductivity
R _{N2H2/GO}	(wt%)	(wt%)	(wt%)	(wt%)	C/N	C/O	(S/m)
0.87:10	60.40	3.53	1.50	34.57	20.1	2.33	4
3.5:10	63.89	3.03	1.17	31.90	24.6	2.67	37
7:10	82.92	3.25	0.11	13.72	19.8	8.06	7222
35:10	85.32	3.15	0.11	11.42	31.6	9.97	7161
70:10	81.25	4.09	0.19	14.46	23.3	7.49	7272
700:10	84.25	3.78	0.40	11.57	26.0	9.71	7287

Table 2.2 Elemental analysis results of graphene prepared at various hydrazine contents [17]

2.4 Graphene suspension in polymer solution

Selection of mixing method is a crucial factor to obtain excellent properties of composite because graphene, graphene oxide and their derivatives are difficult to disperse in polymer matrix to form suspension. There are three methods for preparing polymer composite that are physical mixing, solution b lending and *in situ* polymerization. [19] The physical mix ing is the method in which filler and polymer are mixed together v ia physical forces such as stirring, shear mixing at the molten state of polymer matrix or polymer solution. A lthough t his process is easy and simple, filler c an e asily be

aggregated in matrix. The second method is the solution blending. Filler is suspended in solvent and then mixed together with po lymer and solvent is then evaporated. This process improves the dispersion of filler in polymer but it is difficult tor emove all solvent from composite. The last technique is *in situ* polymerization. In this method, filler is mix ed with monomer or solution of monomer followed by polymerization resulting in well dispersion because monomer is inserted between the layers of filler; however, it is quite complicated. Because plate-like particle of graphene / polymer composite contains a few layers of graphene f rom chemical method, the composite h as three s tates of dispersion i ncluding stacked platelet, i ntercalated platelet and exfoliated platelet. The morphology of polymer composite depends upon the degree of dispersion i n polymer matrix. The state of plate-like particle dispersion in polymer matrix is illus trated in Figure 2.4. [19]

Because the system of graphene s uspended in polymer matrix s hows different rheological properties compared to that of n eat polymer, the flow behavior of graphene suspension must be studied since it is an important factor to design processing condition which will be mentioned in section 2.5.



Stacked platelet

Figure 2.4 The state of plate-like particle dispersion in polymer matrix. [19]

2.5 Rheological properties of platelet particle in polymer solution

The rheological properties of particulate suspensions are sensitive to the size, size distribution, s hape, surface characteristics of the dispersed phase and type of media or continuous p hase. R heology potentially offers w ay to a ssess the state of d ispersion of composites in the solution. [20-22] The r heological properties can be measured in two modes: s teady s tate an d dynamic o r oscillatory measurement. T he s teady state measurement s hows t he characteristic of macroscopic behavior of s uspension such as flow a bility, viscosity a nd modulus of s ystem. The flow a bility of suspension c an be divided into 5 s ystems: N ewtonian, B ingham p lastic, ps eudo pl astic (shear t hinning),

dilatants (shear t hickening) and shear t hinning with yield value. [16, 22, 23] Their rheological models are different in their flow behavior which depend on four factors that are size (size distribution) and shape of particle, balance of interaction force, prehistory of suspension a nd a dditional formation s uch as bridging, w hich shows c omplex rheological behavior. [22, 23] In general, particle suspension often shows shear thinning with yield value so the review is considered only this type of behavior. In particle size and particle s ize d istribution, t he large s ize o f d iscrete ph ase exhibit high value o f viscosity but low yield stress because the yield stress depends on interaction between particles. The small-sized particle has large surface area at the same volume fraction when compared with large-sized particle. The increasing of surface ar ea a ffects aggregation of small size of particle. It occurs easily due to attractive force which is higher than repulsive force. Then, the small-sized particle may be form cluster structure which means t hat t he d iscrete ph ase is d ifficult to f low. [24-27] In t erm o f size distribution e ffect, the b road size d istribution of particle will s how lower apparent viscosity than narrow size d istribution. Bi or multimodal size d istribution exhibits lower apparent viscosity than monomodal size distribution. In this case, the broad distribution and multimodal distribution of particles have much range of particle sizes in the system. The fraction of small particle prefers to stay at the vacant space between large particles and behave as a lubricant thus the viscosity and yield stress of suspension is reduced. [25-31] Regarding to the shape of particle, the particle with high a spect ratio (L/D) or far away from spherical shape such as ellipsoid, fiber, rod and disk shape, would like to receive some extra energy dissipation or rearrange structure during flow so the viscosity increases. [27, 28] The pr e-history of suspension is a nother f actor w hich a ffected rheological properties. Goudoulas and coworker study the effect of pr e-shear time and pre-shear value. They found that the effect of pr e-shear time to apparent viscosity in steady state measurement is negligible but when testing in frequency sweep measurement on viscoelastic regime, the modulus, viscosity and tan delta are shifted.[31]

The platelet particle suspension system often shows shear thinning behavior. The example of t his system which is studied from other researchers is c lay suspension, r ed blood cell system, aluminum oxide suspension and graphene suspension. [20, 21, 28, 32] Although there are many model to explain flow behavior of suspension system, most of them are extended from the general equation that is displayed in equation 2.1.[21]

$$\boldsymbol{\tau}^{m} = \boldsymbol{\sigma}_{\boldsymbol{y}}^{\boldsymbol{m}} + \boldsymbol{\eta}_{pl}^{\boldsymbol{n}} \boldsymbol{\dot{\boldsymbol{y}}}^{\boldsymbol{n}}$$
(2.1)

In this expression, there are three functional parameters that affect flow behavior which are effective viscosity (η), shear stress (σ), and shear rate (p). The ultimate shear stress (σ_y) is the maximum value of system that behaves as e lastic solid. It means that solid state transform to fluid state at $\sigma > \sigma_y$. At higher ultimate shear stress or yield point, the interaction of dispersed phase or interaction between dispersed phase and continuous phase is broken do wn thus the composite is de formed and begins to flow. The plastic viscosity (η_{pl}) is the infinite viscosity of c omposite w hich flows a fter de formation and destruction of a ggregation from c olliding of particle. It shows the equilibrium state of dispersed system after shear flow.[21-23] . There are many model at present to explain shear thinning behavior of suspension and composite system, for example Hershell-Bulkey model, Sisko model, Bingham model, Casson model and Cross model which are displayed in equation 2.2-2.6 below.[22] The different model c an be used to explain the different behavior of flow deformation and it can be used to group the material for using in the future application.

Hershell-Bulkey model: $\tau = \tau_{\beta} + k \dot{j}^{n}$ (2.2)

Sisko model:
$$\eta = \eta_{\infty} + \frac{K_2}{\dot{\gamma}}$$
 (2.3)

Bingham model: $\boldsymbol{\tau} = + \boldsymbol{\eta}_{pl} \boldsymbol{\dot{\gamma}}$ (2.4)

Casson model:
$$\tau^{1/2} = \tau_c^{1/2} + \eta_c^{1/2} \dot{\gamma}^{1/2}$$
 (2.5)

Cross model:
$$\frac{\eta - \eta(\infty)}{\eta(0) - \eta(\infty)} = \frac{1}{1 + K \dot{\gamma}^m}$$
(2.6)

Another mode of rheological measurement is dynamic or oscillation measurement which can demonstrate a characteristic of microstructure or interaction between particles such as interaction force, network structure and flocculation formation. In this mode, the rheological properties can be determined from strain sweep, frequency sweep and time sweep measurement. L inear viscoelastic r egion can be de tected from s train sweep measurement. T he point w here structure of suspension breaks do wn w hich can be indicated by a starting point that storage modulus (G') begins to de crease. [22, 33] Characteristic of discrete phase such as size and shape and the type of continuous phase or matrix of system are major contribution to strain in which the non-linear viscoelastic begins. If the matrix be haves as e lastic material, the linear vi scoelastic region is eas y observed but the polymer's olution's ystem's how opposite behavior. [20, 33-35] The polymer solution generally displayed viscous character so high enough content of particle or di screte phase must be add ed to b e a ble t o observe liner viscoelastic r egion o f suspension system. The frequency sweep measurement gives the information regarding how composite behaves r egarding e lastic a nd viscous pr operties at d ifferent concentration. In this test the behavior of system depends on character of discrete phase and continuous phase. The changing behavior can be found from cross-over point where storage modulus, which is the characteristic of elastic solid, shows the same values loss modulus, which is the characteristic of viscous liquid. [33-38] The cross-over point could be found if the continuous phase has elastic solid character or dispersed phase has high concentration enough or modification of surface particle with surfactant. The surfactant affects interaction between polymer matrix and particle which may be formed network structure. [20, 38] In the system of particle suspension in polymer solution, the cross-over point is hardly observed because polymer matrix shows viscous characteristic and at the higher concentration of particle, the system will be changed to another system from the beginning such as paste material. The time s weep measurement is the method for studying stability of polymer and polymer composite at constant stain and frequency which r elates t o s uitable a nnealing t ime and structure of c omposite for us ing in processing. For example, K im a nd c oworker studied t he stability of gr aphite a nd graphene in polycarbonate composite [37], w hereas Wang a nd c oworker studied t he stability of organoclay in polylactic acid composite. [34]

CHAPTER III

EXPERIMENTS

3.1 Chemicals

In this research, graphite (99.99% purity, particle size ≤ 45 micron), hydrazine hydrate (55.1 w t%) and poly [(phenyl g lycidyl e ther)-co-formaldehyde] whose number range molecular weight (M_n) was about 345 g/mol, EEW = 174.8, viscosity 1595 centipoises at 52 °C were purchased from S igma Aldrich to us e as raw material, reducing agent for graphene synthesis and epoxy solution as media respectively. The other chemicals were purchased from A jex F inechem that were sulfuric ac id solution (96.3 wt%), potassium permanganate (100% purity), sodium nitrate powder (99.69% purity), sodium hydroxide pellet (99.65% pur ity), and hydrogen peroxide (30 v ol%). Hydrochloric a cid fuming (37.6 vol%) purchasing from Merck was used for washing metal ion after graphene oxide synthesis. Finally dimethylformamide (99.8% purity) purchasing from RCI labscan was used as solvent for dispersed particle.

3.2 Experimental method

3.2.1 Synthesis of graphene oxide

Graphene oxide was synthesized from graphite by modified Hummers method. 2 g of graphite powder was mixed with 50 ml of concentrated sulfuric acid and 1 g of sodium nitrate powder. After that, 6 g of potassium permanganate powder was slowly added into solution in ice bath for 1 hour and stirred on hot plate at 35 °C for 2 hours. Then 100 ml DI water was poured into the solution and it was heated up to 60 °C. Hydrogen peroxide (30 wt%) was dropped into the solution until colour was changed to yellow brown. Then, 500 ml of d ilute HCl (5 vol%) was poured into it. The solution was stirred and washed with DI water several times until neutral and freeze-dried for collecting powder. The over view process can be seen in chart below.



3.2.2 Chemical Reduction of graphene oxide

Graphene oxide powder was re-dispersed in DI water and adjusting pH with HCl and NaOH solution. The concentration of particle in solution was 1 mg/ml. The pH of solution was a djusted until stable su spension was obtained. The solution was measured via zeta-potential technique and physical observation. The homogenous suspension was chosen to reduce graphene oxide with hydrazine hydrate (5.51 wt% in aqueous solution) at the ratio by weight of hydrazine:GO equals to 7:10. The solution in reflux system was heated-up in oil bath at 90 °C for 10 hours and it was then filtrated and washed with DI water until neutral and freeze-dried for collecting powder.

3.2.3 Sample preparation for rheological testing

Graphene was mixed in poly [(phenyl glycidyl ether)-co-formaldehyde] by magnetic stirring and sonication at 0-9 wt% of graphene. Rheological properties were studied via cone and plate rheometer. Before beginning of the test, the samples were pre-shear at 0.1 s⁻¹ of s hear rate; 30 sec for continuous ramp shear rate mode and pre-shear at 1 s⁻¹ of shear rate; 30 sec for oscillation test.
3.3Characterizations

3.3.1 Atomic force microscopy (AFM)

The sample powder in dimethylformamide was prepared by sonication for 30 minutes, coated on Si substrate and dried in oven overnight. After that, it was put on the stack. The sample and tip were then inserted into the microscope, aligned, adjusted maximum energy and scaned sample. The sample was tested in non-contact mode, using Nchr cantilever and collecting data with topology, non-contact mode (NCM) a mplitude and non-contact mode (NCM) phase.

3.3.2 Fourier Transform Infrared Spectroscopy (FTIR)

FTIR was used to identify functional group of graphene product. It was performed in the range of 4000-650 cm⁻¹, resolution 4 cm⁻¹ and identified peak in absorbance mode. The potassium bromide (KBr) was used as background material. Powder was milled with KBr and compressed by hydraulic press at 10 tons for 1 minute and put into the sample holder. The holder was placed into the sample chamber and the spectrum was recorded at least 64 scans for each spectrum.

3.3.3 Thermal Gravimetric analysis (TGA)

The degradation temperature of material was tested by TGA under nitrogen. Powder s ample (7-10 mg) w ere loaded in c eramic pa n and r amped up t o 800 °C a t 5 °C/min of heating rate. The N₂ gas flow rate of system was set at 50 ml/min.

3.3.4 Scanning Electron Microscopy (SEM) and Transmission Electron Microscopy (TEM)

The sample powder was dispersed in dimethylformamide and sonicated for 30 minutes, coated on Si substrate and dried in oven overnight. After that, it was put on the stack that was attached with carbon tape. It was coated with gold for conduction and put into the chamber.

3.3.5 Raman Spectroscopy

Sample powder was dried in oven overnight and pressed into the sample holder. Nd-YAG was used as the source of laser at wavelength 1064 nm. The power was applied about 50 mW and resolution of this experiment was 16 cm⁻¹.

3.3.6 Zeta sizer (Zeta-potential)

The colloidal particle in aqueous suspension (1 mg/ml) at different pH value was loaded into the cell and the cell was inserted in the chamber. After that, the electric field was applied across the particle suspension. When equilibrium was reached between two opposite charge, the velocity was measured and translated to zeta-potential value.

3.3.7 UV-VIS-NIR Spectroscopy

The colloidal particle in a queous suspension (0.4 mg/ml) at different pH value was loaded into the cell and inserted it in the chamber. The spectrum source was applied to the sample chamber which was black body. The spectrum was scanned from 900 nm to 100 nm and the absorbance of suspension spectrum was recorded.

3.3.8 pH meter

The pH meter was calibrated with standard solution at pH4, pH7 and pH10 before using. T he pH pr obe was washed with d istillated water to remove the electrolyte that must always be soaked when it is not in use. Then, the pH probe was put into sample solution and collected the pH value from display.

3.3.9 Cone and plate rheometer

Graphene was m ixed in poly [(phenyl glycidyl e ther)-co-formaldehyde] at different concentration by magnetic stirring for 30 minutes followed by sonication for 45 minute. Then, it was put on the plate of rheometer and set to geometry gap (Cone angle 2° , diameter 20 and 40 mm). Nitrogen gas pressure was set at 2 bars and air was set at 8 bars. The sample was pre-sheared at 0.1 s⁻¹ for steady state measurement and 1 s⁻¹ for oscillation measurement. The time of pre-shear was 30 second and then, rest the sample for equilibrium at 1 minute before testing the sample.

Table 3.1	Measurement and	characterization	of sample	e
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Measurement	Specification	Characterizations	
Atomic force microscopy	Park system, Model: XE	Surface morphology,	
(AFM)	100, Korea	thickness of particle and	
		particle size distribution	
Fourier Transform	Perkin Elmer instruments,	Functional group of	
Infrared Spectroscopy	Model : Spectrum GX	materials	
(FTIR)	Raman FTIR, USA		
Thermal Gravimetric	Perkin Elmer instruments,	Thermal stability,	
analysis (TGA)	Model: Diamond	Degradation temperature	
	TG/DTA, USA		
Scanning Electron	JEOL, Model : JSM-6400,	Surface morphology	
Microscope (SEM)	Japan		
Transmission Electron	JEOL, Model : JEM-2100,	Surface morphology	
Microscope (TEM)	Japan		
	Perkin Elmer instruments,	Classified structure of	
Raman spectroscopy	Model : Spectrum GX FT-	material	
	Raman , USA		
Zeta sizer	Malvern, Model : ZEN	Surface charge of colloid	
	3600, UK	particle	
pH meter	Exact Instrument, Model :	pH value	
	CT-6020A, China		
UV-VIS spectroscopy	Varian, Model : Carry	Colloid stability	
	5000, Austraria		
Cone and plate rheometer	TA Instruments, Model :	Rheological properties	
	AR G2, USA		

CHAPTER IV

RESULTS AND DISCUSSION

The experimental results from this thesis will be demonstrated and discussed dividing into three parts. The first part concentrates on characterization of particle from synthesis process which is chemical structure, thermal properties, morphology, thickness and diameter of particle. The second part explains the electrostatic stability of graphene oxide in aqueous with various pH. The third part focuses on rheological properties of graphene suspended in epoxy liquid at different concentration of particle to show flow behavior and interaction of particle suspension.

4.1 Particle characterization

4.1.1 Chemical structure analysis

Graphene oxide was synthesized from graphite by modified Hummers method. G raphene w as o btained by chemical r eduction o f gr aphene o xide w ith hydrazine. The functional group of graphite, graphene oxide (GO) and graphene were observed by FTIR as shown in Figure 4.1. The peak around 1700, 1650 and 1250 cm⁻¹ corresponds to carboxyl and carbonyl stretching of graphene oxide respectively. The epoxy peak of graphene oxide was shown at 1000 c m⁻¹ and hydroxyl stretching peak was shown at 3400 cm⁻¹. FTIR spectrum would confirm that GO had epoxy, hydroxyl, carboxyl and ketone group which was described by other r esearcher's work.[1, 7-9] The pe aks a round 1600 -1700 cm⁻¹ and 1150 c m⁻¹ of gr aphene w ere car boxyl a nd carbonyl group respectively. From this result, it can be confirmed that the epoxide and hydroxyl group o f graphene o xide was reduced from hydrazine in which Gao and coworker proposed mechanism o f t his reaction[39]. Although it has many different ways of reaction, the epoxide and hydroxyl group are reduced.



Figure 4.1 FTIR spectra of graphite, graphene oxide and graphene.

4.1.2 Morphology of graphene

Graphite, graphene oxide and graphene in aqueous solution were coated on Si substrate and dried in oven overnight. It was found that the size of graphene oxide and graphene were smaller than graphite because of oxidation reaction via modified Hummer m ethod. D uring o xidation, gr aphite t hat c onsists o f 100 -1000 gr aphene layers was separated and had the oxygen functional group on the surface which was reduced by c hemical r eduction w ith hydrazine t o gr aphene sheet. T he o xygen functional group had an effect on thermal and electrical properties of graphene. The products of synthesis were shown in Figure 4.2 in which the scale bar on SEM image of graphite, graphene oxide and graphene are 10 micrometers. The graphene product was slightly smaller size than graphene oxide particle. Because the chemical reduction just reduced oxygen functional group, it does not separate graphene sheets.



c) SEM image of Graphene

d) TEM image of Graphene



Figure 4.2 Electron microscope image of graphite, graphene oxide, graphene from SEM (a,b,c) and graphene from TEM. (d)



Figure 4.3 Topological image of AFM of graphite, graphene oxide and graphene respectively

Sample	Ferret's diameter (nanometer)	Thickness (nanometer)
Graphite	14,758	1,720
Graphene oxide	712	21
Graphene	561	12

Table 4.1 Result of graphite, graphene oxide and graphene from figure 4.3



Figure 4.4 Particle size distribution of graphene product. (Data was collected from

240 particles)

The atomic force microscopy analysis can support the theory as well as the result from microscopy technique because the thickness of graphite and graphene oxide particle decrease s ignificantly from 1.7 micrometers to 21 nanometers. The thickness of graphene is much thinner than that of graphite due to separation layer by oxidation process. In term of Ferret's diameter, graphene oxide and graphene showed significantly smaller s ize than graphite. G raphene is slightly smaller than graphene oxide w hich may be o ccurred from c ollided p article dur ing c hemical r eduction process. G raphene is slightly thinner than graphene oxide w hich c ould be due to oxygen functional group reduction from hy drazine m echanism at both upper a nd lower plane of graphene sheet.[39] From AFM analysis, it can be estimated that the product had 10-20 layer based on thickness of c arbon atom with oxygen functional group which still remains in the product.[8, 40]

Figure 4.4 showed the particle size distribution from AFM image. The size of graphene product is in the range of 0.2-1.2 micrometers. The arithmetic count mean diameter from size distribution was 0.581 micrometers. The standard deviation of this size distribution was 0.014. The particle size distribution was one factor that a ffects flow behavior of graphene suspension so I would like to explore it.

4.1.3 Raman spectroscopy

Raman spectra of graphite, graphene oxide and graphene were shown in Figure 4.5. Generally, Raman spectrum of car bon material was detected in G and D peak which is the bonding of sp^2 atom in both ring and chain and breathing mode of sp^2 atom in ring of c arbon-carbon atom. Graphene o xide a nd graphene showed two intense features of G peak at about 1600 c m⁻¹ and D peak at about 1300 c m⁻¹ while

graphite had intensity of G peak but it was very low. Graphene oxide and graphene had intensity of G peak at 1612 and 1589 cm⁻¹. Both of graphene oxide and graphene had higher intensity of G pe ak than graphite because s ignificant s tructure changes during chemical process.[41] It can be confirmed that graphene and graphene oxide were the structure in sp² carbon in hexagonal ho neycomb s tructure. The D peak of graphene oxide and graphene are shown at 1320 and 1289 cm⁻¹. The D peak displayed the disorder or amorphous of carbon lattice.[42, 43] It was found that graphene oxide had the intensity of G peak higher than D peak but graphene showed opposite trend because of the reduction of oxygen functional group with hydrazine.[10, 44, 45] The intensity of D peak to G peak ratio of graphene oxide and graphene were 0.98 and 1.36 respectively. The in-plane crystallite size (L_a) can be calculated from intensity ratio I $_D/I_G$ which was the c haracteristic of car bon material t o display the e lectrical behavior. The in-plane crystallite size was computed from L_a = (560/E_{laser})(I_G/I_D). From t his r elation, t he in-plane crystallite size of gr aphene pr oduct w as 220 nanometers. [9, 45, 46]



Figure 4.5 Raman spectrums of graphite, graphene oxide and graphene

4.1.4 Thermal properties of graphene

The t hermal s tability o f gr aphite, gr aphene oxide a nd gr aphene w as investigated via thermal gravimetric analysis under nitrogen atmosphere. The results were shown in Figure 4.6. Graphene oxide showed significant weight loss at 100 and 200 °C which was the evaporation of water and decomposition of oxygen containing group respectively. From the result, graphite and graphene was more thermally stable than graphene oxide but graphite was not used as filler in this research because of its large particle, less e lectrical and thermal conductivity.[1, 3, 37] Graphene was less stable than graphite because of amorphous carbon structure from acid treating during graphene o xide s ynthesis and o xygen functional gr oup r emainder a fter hy drazine reduction.[13, 42, 47]



Figure 4.6 Thermal degradation of graphite, graphene oxide and graphene

(Heating rate 5 °C/min)

4.2 Electrostatic stability of graphene oxide

The zeta-potential of graphene oxide (GO) in aqueous solution at 1 mg/ml was measured via nano sizer instruments at pH 3-pH 14. It was found that graphene oxide was well dispersed in aqueous solution at pH 12 and it became poorly dispersed in aqueous solution at pH higher than 12 whose physical observation are seen in Figure 4.7. The zeta-potential was reduced when the solution had higher pH and it cannot be measured at pH higher than 12 because the zet a-potential can be measured in the system with stable particle in aqueous media. The graphene oxide in suspension at pH higher than 12 precipitated so zeta-potential cannot be measured. From the result of UV-VIS s pectroscopy, it was found t hat gr aphene o xide in aqueous s olution was begin to absorb spectrum around 290 nanometer. The result can be divided into three groups that were the most stable graphene oxide suspension at pH 12 which showed the highest absorbance, the less stable or precipitated graphene oxide suspension (pH 13, pH 14) which showed the lowest absorption of spectrum. At pH 13 and pH 14, graphene oxide rapidly aggr egated and settled down at the bottom because this condition had high amount of sodium ion and hydroxide ion which may reduce the interfacial i nteraction b etween particles and formed flocculation which a ffected colloid stability. [16] T he last group was graphene o xide in a queous suspension whose pH lowers than 12. This group showed lower absorbance than graphene oxide in aqueous suspension at pH 12 but higher than pH 13, pH 14 which had absorption in the range of 0.8-1.8. From above reason, it was supported that at pH 12 graphene oxide in aqueous solution shows the most stable condition from electrostatic repulsive charge method and suitable to use in the step of chemical reduction with hydrazine.



Figure 4.7 Physical observation of GO in aqueous suspension with various pH.



Figure 4.8 The zeta-potential of GO in aqueous suspension at pH 3-14.



Figure 4.9 UV-VIS spectrum of graphene oxide in aqueous media at different pH value.

4.3 Rheological p roperties o f g raphene i n pol y [(phenyl g lycidyl e ther)-coformaldehyde] media

4.3.1 Steady state measurement

The s teady state m easurement o fr heology was s tudied to ex plain the macroscopic s tructure o f gr aphene suspension in t erm o f yield value a nd plastic viscosity. T his t est w as used for ex ploring deformation c haracter and processing parameter o f material. I n o ur r esearch, graphene w as mixed with po ly[(phenyl glycidyl ether)-co-formaldehyde], epoxy liquid by sonication (180W, 40 kHz) for 45 minutes. The particle suspension was loaded on 25 mm, 2° cone and plate rheometer at 26 °C. The sample was pre-sheared at 0.1 s⁻¹ of shear rate for 30 second and at rest

for equilibrium for 1 minute before continuous ramp of shear rate testing. The result in figure 4.10 shows Newtonian be havior for neat epoxy and become shear thinning behavior when the a mount of gr aphene in epoxy i ncreased. The shear t hinning behavior of high gr aphene loading suspension showed high vi scosity at 1 ow shear rate, the vi scosity reduced when s hear rate i ncreased and became constant at high shear rate. The reason of this behavior was the broken of network structure between particle a nd media or particle interaction at the b eginning thus the r eduction of viscosity d isappeared. When particle and network's structure was broken do wn t o smallest size and can flow, the viscosity of suspension became constant.[22] There are many models for characterization of particle suspension that shows shear thinning behavior s uch as the H erschel-Bulkley G eneral M odel, the cr oss equa tion, S isko model and Casson model [22]. When we plot between $p^{1/2}$ and $\tau^{1/2}$, it showed linear behavior which means that the graphene suspension in this epoxy fitted Casson model which can be seen in Figure 4.11.



Figure 4.10 The continuous ramp shear rate of graphene suspension in poly [(phenyl glycidyl ether)-co-formaldehyde] via cone and plate (cone angle 2°) rheometer.



Figure 4.11 Plot between $r^{1/2}$ and $\tau^{1/2}$ from 3 wt% graphene suspension in poly [(phenyl glycidyl ether)-co-formaldehyde].

The C asson model is the flow c urves u se for many paints and suspension system whose equation is shown as follows

$$\tau^{1/2} = \tau_c^{1/2} + \eta_c^{1/2} \, \mathbf{\dot{r}}^{1/2} \tag{4.1}$$

Where $\tau_c = Casson yield value$

 η_c = Casson plastic viscosity



Figure 4.12 Casson yield value at different concentration of graphene suspension in poly[(phenyl glycidyl ether)-co-formaldehyde].

From equation 4.1 and graph in figure 4.11, the τ_c and η_c can be calculated from intercept a nd s lope o f gr aph t hat a re 4. 427 P a and 81. 559 P a.s r espectively. The Casson yield value (τ_c) is the shear stress which is used to exert the initial flow and deformation o f composite. Moreover it can inform the degree of aggregation in the system. From figure 4.12, the yield value of graphene suspension at 1 and 2 wt % had significant lower than 3 wt% graphene suspension because low particle loading does not obstruct flow of suspension thus t he s uspension can flow easily which c an be displayed from low yield stress. The Casson plastic viscosity (η_c) is a function of shear stress which must be used to maintain constant flow when aggregate of particle was destructed to smaller size by shear force in dispersed medium.[21] It can indicate how well flow ability of composite is. From this result, it can be used to predict flow behavior of gr aphene in suspension system which has the same characteristic for future application.

4.3.2 Oscillation measurement

The oscillatory measurement of rheology was tested to identify microstructure of material and classified suspension material. From this mode, it can be explained how the behavior of material is, for example; how much the material shows elastic solid character and viscous liquid character. The elastic solid and viscous liquid character is the useful for processing. Moreover, the oscillation test can be used to assess t he d egree o f d ispersion o f s uspension in viscous media. I n t his research, graphene was mixed with epoxy by sonication for 45 m inutes. The suspension was loaded on 40 mm, 2° cone and plate rheometer at 26 °C. The sample was pre-sheared at 1 s⁻¹ of s hear rate for 30 second and at rest for equilibrium for 1 minute before oscillatory testing. The s train sweep measurement w as us ed to obtain linear viscoelastic r egion. T he linear viscoelastic r egion is o bserved when t he s torage modulus do es not c hange w hen s train is applied.[22] It m eans t hat t he s torage modulus is constant with varied strain. In this region, the structure of suspension can be r econstructed to or iginal shape when shear f orce is r emoved. Because s torage modulus is the characteristic of elastic solid and the epoxy liquid showed absolutely viscous character so storage modulus of neat epoxy was hardly detected. From this reason, the storage moduli of ne at epoxy and suspensions at lower than 5 wt% of graphene whose storage modulus was low and cannot see linear viscoelastic region,

were not shown in figure 4.13. It was found that at 5 w t% of graphene suspension; linear viscoelastic r egime was o bserved from storage m odulus a nd was shown obviously when the concentration of particle increased. However, the system with up to 9 wt% of graphene was investigated because suspension with higher than 9 wt% of graphene it became paste material which was not in scope of this work. The graphene suspension in this epoxy showed critical strain at 0.001 strains. The critical strain was the point whose storage modulus begins to decrease significantly. At this point, the structure or physical bonding of aggregated particle started to break down.[21, 22] In frequency sweep measurement, the strain was set at 0.001 and frequency was varied from 1 rad/sec to 628 rad/sec. From figure 4.14, the storage modulus of suspension increased when the particle's concentration increased. In frequency sweep test, the slopes of storage modulus at 8-9 wt % of graphene decreased when concentration of graphene increased which can be explained that the suspension showed pseudosolidlike behavior. The storage modulus of suspension increased as increasing frequency displayed more solid-like behavior at high frequency. In term of loss modulus (Figure 4.15), the results can be divided into three groups that were neat epoxy which showed the lowest v alue, 5 -7 wt% gr aphene in s uspension w hose value w ere t he s ame behavior a nd 8-9 wt% g raphene s ystem w hose loss moduli increased at higher concentration of particle in suspension. In the group of 8-9 wt% graphene system, the higher concentration of particle indicated higher dissipation energy because in this region graphene suspension had high concentration of particle so suspension hardly moved because particle obstructed the flow of suspension. For 5-7 wt% system, the value of loss moduli was not changed although the concentration of particle increased. It can be explained that at this condition the suspension was diluted and discrete phase

was easy to move with continuous phase thus it did not a ffect dissipation energy. When compared between the effect of graphene loading in term of storage modulus and loss modulus it was observed that the storage modulus increased greater than loss modulus which means that graphene loading had more effect on elastic solid character than viscous character.[48] Figure 4.16 shows the pseudosolid like b ehavior from frequency sweep measurement. For 8 and 9 wt% graphene system, phase angle was low at small frequency and increasing at high frequency which mean that at low frequency, 8 and 9 wt% of graphene showed higher elastic solid character than high frequency. While the phase angle of 5-7 wt% graphene system and neat epo xy was not changed with various angular frequencies. The phase angle of them was close to 90 degree which was the character of viscous material. [22, 23, 48] Finally, when plotting data between complex modulus (G*) and phase angle in figure 4.17 it was found t hat at 8 a nd 9 wt% of graphene system, t he phase a ngle was low at small complex modulus and increasing when complex modulus increased. This behavior relates t o ps eudosolid-like because gr aphene can form network structure or agglomeration. This character can be found if concentration of particle in suspension system was high enough thus it was not observed in 5-7 wt% of graphene suspension which was dilute system. [49]



Figure 4.13 The storage modulus results of graphene suspension from stain sweep measurement. (at 1 rad/sec)



Figure 4.14 The storage modulus results of graphene suspension from frequency

sweep measurement.



Figure 4.15 The loss modulus results of graphene suspension from frequency sweep measurement.



Figure 4.16 The phase angle of graphene suspension in term of angular frequency from frequency sweep measurement.



Figure 4.17 The phase angle of graphene suspension in term of complex modulus from frequency sweep measurement.

CHAPTER V

CONCLUSION AND RECOMMENDATION

5.1 Conclusion

In the first part, the particle from synthesis was characterized by FTIR, SEM, TEM, A FM, R aman sp ectroscopy a nd T GA t o i dentify properties o f gr aphite, graphene oxide and graphene product. It was found from chemical structure analysis that graphene product still had carboxyl and carbonyl group after chemical reduction with hy drazine. From morphology a nalysis, graphene o xide and graphene had the proximity size because c hemical r eduction d id not s eparate gr aphene s heet w hile graphite w as larger t han graphene o xide due t o separation s heet from o xidation process. T he s ynthesized graphene had the ar ithmetic co unt m ean d iameter f rom particle size distribution around 581 nanometers and standard deviation around 0.014. The thickness of graphene product was 12 nanometers in which the number of layer could be around 10-20 layers based on c arbon bonding of he xagonal ho neycomb structure added with inter-layer spacing and some oxygen functional group remained at upper and lower planes. Then, the intensity of Raman spectrum was u sed to calculate in-plane crystallites ize o f gr aphene pr oduct w hich w as a bout 220 nanometers. Finally, thermal stability of graphite, graphene oxide, and graphene were tested by T GA. I t w as o bserved that gr aphene w as more thermally stable than graphene oxide because of oxygen functional group reduction but it was less stable than gr aphite because oxygen f unctional gr oup remainder from s ynthesis and amorphous carbon from acid treating during oxidation process.

In the second part, graphene o xide in aqueous suspension was a nalyzed in term of colloidal stability at different pH for up-scale process in the future. It was observed that at pH 12 graphene oxide showed the most stable colloidal suspension in water which was confirmed by physical observation, zet a-potential t echnique and UV-VIS spectroscopy.

Finally, rheological properties of graphene product was studied to explain the flow b ehavior of s uspension and interaction of particle in poly [(phenyl g lycidyl ether)-co-formaldehyde], e poxy liquid. It was o bserved that graphene suspended in epoxy displayed shear thinning b ehavior when i ncreasing shear r ate. T he flow properties of this suspension can be fitted in Casson model in which the Casson yield value ($\tau_c = 4.427 \text{ P a}$) and Casson p lastic viscosity ($\eta_c = 81.559 \text{ P a.s}$) can be calculated from intercept and slope of gr aph between $r^{1/2}$ and $\tau^{1/2}$. T he line ar viscoelastic region of graphene suspended in e poxy was at strain be low 0.001 and began to observe linear viscoelastic solid region from storage modulus at 5 wt% of graphene content. The physical bonding of particle-polymer or particle-particle breakdown can be indicated at strain above 0.001. The linear viscoelastic properties were measured by frequency sweep from 1 rad/sec to 628 rad/sec. It was found that the storage modulus increased greater than loss modulus at the same concentration of graphene p article which m eans that graphene loading had more effect on s torage modulus than loss modulus. The pseudosolid-like behavior showed obviously at 8-9 wt% of graphene which nearly became to paste material.

Recommendations

The s ynthesized step should be o ptimized such as r eaction t ime at c ool temperature during graphene oxide synthesis by modified Hummer method because it may affect the separation of graphene layer process and the reaction time for chemical reduction with hydrazine should be varied too because it affects oxygen content of graphene product. Moreover, the freeze-dry process after oxidation process to collect graphene oxide powder must be omitted to avoid particle agglomeration which affects thickness of graphene product. The filtration process after chemical reduction must be changed t o another process for protecting the p article agg lomeration via p hysical bonding of particle-particle interaction during filtration process.

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APPENDICES

Appendix A

Calculation of hydrazine to graphene oxide ratio

- Using hydrazine hydrate 5.51 wt% (dilution from 55.1 wt% in aqueous, density = 1.029 g/cm³)
- Synthesis graphene from graphene oxide 400 mg/100 ml of solution/batch
- Ratio of hydrazine : graphene oxide = 7:10

Using hydrazine (5.51 wt%) = $\frac{100 \text{ x } 70 \text{ x } 4}{5.51 \text{x} 1.029 \text{x} 1000}$ = 4.938 cm³

Appendix B

Calculation of In-plane crystallite from Raman spectrum

Calculate from laser energy

- Using laser energy of Nd:YAG = 1.17 eV
- I_D/I_G of graphene spectrum was 1.356

$$L_{a}(nm) = \frac{560}{E_{laser}^{4}} \left(\frac{I_{D}}{I_{G}}\right)^{-1}$$
(1)

From equation (1)

La (nm) =
$$\frac{560}{(1.17)^4 (1.356)}$$

Calculate from wave length

• Wave length of Na:YAG laser was 1064 nm

$$L_{a}(nm) = (2.4 \times 10^{-10}) \lambda_{laser}^{4} (\frac{I_{D}}{I_{G}})^{-1}$$
 (2)

From equation (2)

La (nm) =
$$\frac{(2.4 \times 10^{-10}) (1064)^4}{(1.356)}$$

= 226 nm

From both equations can be calculated in-plan crystallite size about 220 nm

Appendix C

Data of particle size distribution and calculation of particle diameter

										$(D_i -$	$f(D_i)$ -
D_{uper}	D_{lower}	ΔD_i	D_i	n_i	f_{ni}	$f_{ni}/\Delta D_i$	C_{ni}	$f_{ni}D_i$	D_i - D_a	$(D_a)^2$	$(D_a)^2$
0.20	0.25	0.05	0.225	2	0.008	0.167	0.008	0.002	-0.356	0.13	0.001
0.25	0.30	0.05	0.275	10	0.042	0.833	0.050	0.011	-0.306	0.09	0.004
0.30	0.35	0.05	0.325	18	0.075	1.500	0.125	0.024	-0.256	0.07	0.005
0.35	0.40	0.05	0.375	21	0.088	1.750	0.213	0.033	-0.206	0.04	0.004
0.40	0.45	0.05	0.425	27	0.113	2.250	0.325	0.048	-0.156	0.02	0.003
0.45	0.50	0.05	0.475	28	0.117	2.333	0.442	0.055	-0.106	0.01	0.001
0.50	0.55	0.05	0.525	24	0.100	2.000	0.542	0.053	-0.056	0.00	0.000
0.55	0.60	0.05	0.575	18	0.075	1.500	0.617	0.043	-0.006	0.00	0.000
0.60	0.65	0.05	0.625	15	0.063	1.250	0.679	0.039	0.044	0.00	0.000
0.65	0.70	0.05	0.675	13	0.054	1.083	0.733	0.037	0.094	0.01	0.000
0.70	0.75	0.05	0.725	11	0.046	0.917	0.779	0.033	0.144	0.02	0.001
0.75	0.80	0.05	0.775	10	0.042	0.833	0.821	0.032	0.194	0.04	0.002
0.8	0.85	0.05	0.825	9	0.038	0.750	0.858	0.031	0.244	0.06	0.002
0.85	0.90	0.05	0.875	7	0.029	0.583	0.888	0.026	0.294	0.09	0.003
0.90	0.95	0.05	0.925	6	0.025	0.500	0.913	0.023	0.344	0.12	0.003
0.95	1.00	0.05	0.975	6	0.025	0.500	0.938	0.024	0.394	0.15	0.004
1.00	1.05	0.05	1.025	6	0.025	0.500	0.963	0.026	0.444	0.20	0.005
1.05	1.10	0.05	1.075	5	0.021	0.417	0.983	0.022	0.494	0.24	0.005
1.10	1.15	0.05	1.125	3	0.013	0.250	0.996	0.014	0.544	0.30	0.004
1.15	1.20	0.05	1.175	1	0.004	0.083	1.000	0.005	0.594	0.35	0.001
				240				0.581			0.0478

Table C.1 Data sheet of graphene particle size calculation

Standard deviation Calculation

S.D. =
$$\sqrt{\frac{\epsilon f (Di - Da)^2}{N}}$$

S.D. = $\sqrt{\frac{0.0478}{240}} = 0.014$

VITA

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